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NASA Technical Memorandum 81485

(NASA-TM-81485) EPPECIS OF YITRIUM, N60-22464
ALUMINUM AND CEROMIUM CONCENTRATIONS IN BOND
COATINGS ON THE PERFORMANCE OF
ZIRCONIA-YITRIA THERMAL BARRIERS (NASA) Unclas
12 p HC AC2/MF A01 CSCL 11F G3/26 17998

CHROMIUM CONCENTRATIONS IN BOND COATINGS ON THE PERFORMANCE OF ZIRCONIA-YTTRIA THERMAL BARRIERS

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Prepared for the International Conference on Metallurgical Coatings sponsored by the American Vacuum Society San Diego, California, April 21-25, 1980

EFFECTS OF YTTRIUM, ALUMINUM AND CHROMIUM CONCENTRATIONS IN BOND COATINGS ON THE PERFORMANCE OF ZIRCONIA-YTTRIA THERMAL BARRIERS

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ABSTRACT

A cyclic furnace study was conducted on thermal barrier systems to evaluate the effects of yttrium, chromium, and aluminum in nickel-base alloy bond coatings and the effect of bond coating thickness on yttria-stabilized zirconia thermal barrier coating life. Without yttrium in the bond coatings, the zirconia coatings failed very rapidly. Increasing chromium and aluminum in the Ni-Cr-Al-Y bond coatings increased total coating life. This effect was not as great as that due to yttrium. Increased bond coat thickness was also found to increase life.

INTRODUCTION

Significant progress has been made in improving the performance and adherence of two-layer thermal barrier systems (TBS) on metallic substrates $^{1-4}$. The reported Lewis Research Center data indicate that the performance of the NiCrAlY/ZrO2-Y2O3 duplex system is best at certain specific NiCrAl-Y bond coating and ZrO2-Y2O3 (TBC) compositions. One of the better systems, Ni-17.0 Cr-5.4Al-0.35Y/ZrO2-7.9Y2O3*, has withstood 2000 l-hour cycles in a Mach 1.0 burner rig at a 1470 °C surface temperature.

Inis study was conducted to examine the effects of yttrium, chromium, and aluminum in the bond coating and the effect of bond coating thickness on the life of $ZrO_2-Y_2O_3$. The evaluations were done in a cyclic furnace.

EXPERIMENTAL

The compositions of the bond coatings and $2r0_2-Y_20_3$ are given in tables I-V. Substrates were cast nickel-base alloys, B-1900+Hf and MAR-M-200+Hf. Flat specimens with all corners and edges rounded were used. The plasma spray depositions of the bond coatings and $2r0_2-Y_20_3$ was done in the same manner as described in references 1 and 4. Coating thicknesses reported in the tables are maximums. Bond coating thicknesses on the specimens used in the evaluation of the effect of bond coating thickness on life were measured with a micrometer. These measurements showed that thickness could vary by + 0.004 cm.

Evaluations were done in a furnace cycled between either 990° and 280° C or 1095° and 280° C. The cycle consisted of a 6 minute heat-up, 60 minutes at

^{*} All compositions in weight percent.

temperature, and 60 minutes of cooling to about 280°C. Specimens were inspected every 12 cycles until failure, denoted by a visible crack, occurred in the IBC. Cyclic furnace tests appear to be reliable for comparison purposes. From prior investigations, it was found that the IBS having the longest life in the cyclic furnace tests also had the longest life in cyclic natural gasoxygen torch rig tests as well as in cyclic Mach 1.0 burner rig tests 1,2,4.

RESULTS AND DISCUSSION

Effect of yttrium-free Ni-Cr, Ni-Al and NiCrAl bond coatings - The data in table I indicate that the use of an yttrium-free bond coating (8.9 micrometers rms) slightly improves the life of the yttria-stabilized zirconia TBC compared to the case with no bond coating, where cleaned substrate surfaces were relatively smooth (2.5 micrometers rms). If mechanical forces, due to interlocking, play the major role in determining IBS life, then the presence of a bond coating should result in a significant increase in life and the three bond coatings in table I should have similar lives. This is not the case. The greatest changes result from changes in bond coating composition. However, considerable roughness is required to achieve adherence of the oxide layer⁵. Thus, chemical-electrostatic bonding and mechanical forces responsible for adherence can act through a larger area. Failure of 2r02-7203 applied directly on B-1900+Hf occurred by separation of the IBC from the substrate. Many small cracks in the 2r02-7.87203 and 2r02-11.57203 TBCs oriented either parallel or at acute angles to the substrate were observed (Fig. 1).

Effect of yttrium on NiCr and NiAl bond coatings - The data in tables II and III show that the addition of yttrium had a very signifiant beneficial effect on IBS life. Additions of 0.52Y improved life many times while 1.52Y additions were only slightly beneficial. Substitution of chromium for aluminum does not affect the life of IBCs as much as does the addition of yttrium.

Metallographic studies of failed $ZrO_2-7.8Y_2O_3$ coated specimens having yttrium-free bond coatings revealed failure at the substrate-bond coating interface and complete bond oxidation at $990^{\circ}C$. The Ni-19.8Cr-0.53Y bond coating after longer exposure at $990^{\circ}C$ was less extensively oxidized. However, the Ni-19.8 Cr-0.53Y bond coating underwent more oxidation at $1095^{\circ}C$ (Fig. 2a) than did the Ni-19.3Al-0.52Y bond coating (Fig. 2b). With yttrium in the bond coatings failure typically initiated through formation of cracks within the IBC near the bond coating, as shown in figure 2, and extension of these internal cracks to the surface. The oxide stringers formed within the as-deposited Ni-19.3 Al-0.52Y bond coating were identified to be principally aluminum oxides. These stringers grew during testing. Electron microprobe data also suggested that yttrium in the bond coating is diffusing toward the bond coating-IBC interface.

Effect of bond coating Cr and Al content - The data in table IV show that Cr and Al also have a significant effect on IBS life. At 990°C, all but the $2r0_2-17.4720_3$ IBCs withstood 1500 l-hour cycles. At 1095°C, the best bond coating was Ni-25.7Cr-5.6Al-0.32Y, which when coupled with $2r0_2-7.8720_3$ gave more than twice the life of a Ni-16.4Cr-5.1Al-0.15Y/ $2r0_2-7.8720_3$ system.

Figure 3 shows Ni-25.7Cr-5.6Al-0.32Y before testing. It is representative of other bond coatings sprayed in air in that it contains oxide particles and

stringers at particle boundaries, is quite rough and convoluted, and is not of uniform thickness. Figure 4 shows the degree of oxidation of the bond coating in the Ni-25.7cr-5.6Al-0.32Y/2 $r0_2$ -7.8Y $_20_3$ system after testing as 990° and 1095°C. Metallography indicated that oxidation resistance decreased from Ni-25.7cr-5.6Al-0.32Y, to Ni-19.9cr-19.2Al-0.33Y, to Ni-16.6cr-10.6Al-0.33Y as did the life of the IBS. Failure of the oxides with these bond coatings originated within the IBC near the bond coating-IBC interface.

Effect of bond coating thickness – The data in table V show that the thicker the bond coating, the longer the TBS life. Uniformity of bond coating thickness is probably also important because a thin spot could lead to a local, premature failure. The data indicate that manually applied bond coatings should be at least 0.015 to 0.020 cm thick. The bond coating thickness values reported in table V indicate the ranges measured with a micrometer before the oxide was applied. Ihinner areas were observed by metallographic examinations. Photomicrographs of the NiCrAl-0.32Y/Zr02-7.8Y203 TBS showed that the 0.003 to 0.007 cm thick bond coating, which failed after 1419 1-hour cycles at 990°C, was nearly completely oxidized. Metallographic examinations of specimens tested at 990°C showed that as bond coating thickness increased, the thickness of the unoxidized bond coating next to the substrate also increased. Similar trends were noted after 1095°C tests.

The effect of bond coating thickness on life is even more strikingly shown by results obtained at $1095\,^\circ$ C with 0.020 cm thick Ni-25.7Cr-5.6Al-0.32Y bond coatings coupled with $7.8Y_2O_3$ — and $6.1Y_2O_3$ —stabilized zirconia coatings. The former withstood 635 and 656 cycles and the latter withstood 668 and 681 l-hour cycles. This is about a 2.5-fold increase in life resulting from an increase in bond coating thickness from about 0.011 to about 0.020 cm. Bond coating thickness did not affect the failure mechanism.

SUMMARY OF RESULTS AND CONCLUSIONS

Bond coating resistance to oxidation has a significant effect on the adherence and life of the IBS. The presence of yttrium in the bond coating is very critical. Without it, the IBS fails rapidly at the substrate-bond coating interface. With yttrium, failure occurs within the oxide coating near the bond coating-oxide coating interface. Yttrium, aluminum, and/or chromium in the bond coating critically affect IBS life. The best bond coating was Ni-25.7Cr-5.6AlO.32Y which was about 2 times better than Ni-16.4Cr-5.1 Al-0.15Y. Coating life improvements of about 1.5 to 2.0 times were obtained with bond coatings exceeding 0.010-0.012 cm.

REFERENCES

- S. Stecura, NASA TM X-3425, (1976).
- 2. S. Stecura, Am. Ceram. Soc. Bull. 56, 1082, (1977).
- 3. S. Stecura and C. H. Liebert, U.S. Patent No. 4,055,705, (1977).
- 4. S. Stecura, NASA TM X-78976, (1978).
- 5. R. C. Tucker, T. A. Taylor, and M. H. Weatherly, <u>Third Conference on Gas Turbine Materials in a Marine Environment</u>, (Bath Univ., England, 1976), <u>Session VII</u>, Paper 2.

TABLE II. - EFFECT OF YTTRIUM CONCENTRATION IN NI-CT BOND COATINGS
ON YTTRIA-STABILIZED ZIRCONIA THERMAL BARRIER COATING LIFE
[Cyclic furnace lest results]

TABLE I. - THE EFFECTS OF VARIOUS-YITRIUM-FREE BOND COATINGS
ON THE PERFORMANCE OF THE YTTRIA-STABILIZED
ZIRCONIA THERMAL BARRIER COATINGS

[Cyclic furnace test results]

Bond coating	Bui	Thermal barrier coating	r coating	Average number
Composition.	Thickness, cm	Composition, wt R	Thickness, cm	of cycles to failure for two specimens
		390° - 250° C		
No bond coating	1	ZrO ₂ -7.4Y ₂ O ₃	0.055	85
Ni-19.8Al	6,000	-7. HY203	.042	103
-20.2Cr	010	-7. *Y,O3	.034	205
-16.2Cr-5.5Al	.01	$-7.8Y_2^{\circ}O_3^{\circ}$	900.	455
No bond coating		ZrO,-11.5Y,O3	.044	35
Ni-19.6Al	.010	-11.5Y2O3	H:0.	3
-20.2Cr	110	-11.5Y,O3	.039	118
-16.2Cr-5.5Al	110.	-11.5Y2O3	.040	326

Composition, T				
32 1.M	Thickness, cm	Composition, wt E	Thickness, cm	ol cycles to failure for two specimens
		990° - 280° C		
Ni-20.2Cr	0.010	ZrO,-7.8Y,O,	9.034	205
-19.5Cr-1.53Y	600	-7.8Y,O3	.042	300
-19.6Cr-0.53Y	.011	-7.8Y2O3	040	1550
Ni-20.2Cr	.613	ZrO,-11.5Y,O,	.03	113
-19.5Cr-1.53Y	140.	-11.5Y,03	040	169
-19.8Cr-0.53Y	.012	-11.5Y2O3	.039	1537
Ni-20.2Cr	.010	ZrO ₂ -17.4Y ₂ O ₂	.037	8
-19.5Cr-1.53Y	600	-17.4Y203	£0.	152
-19.8Cr-0.53Y	.013	-17.4Y203	.040	612
NI-20.2Cr	110.	ZrO,-24.4Y,O,	.042	75
-19.5Cr-1.53Y	110.	-24.4Y203	.046	93
-19.8Cr-0.53Y	.011	-24.4Y2O3	8	38
		1095 ⁰ - 280 ⁰ C		
Ni-20.2Cr	0.012	ZrO,-7.8Y,O,	0.043	и
-19.5Cr-1.53Y	010	-7.8Y203	.043	4
-19.8Cr-0.53Y	010	-7.8Y ₂ O ₃	8 0.	8
Ni-20.2Cr	110.	ZrO ₂ -11.5Y ₂ O ₃	.043	13
-19.5Cr-1.53Y	110.	-11.5Y203	.038	77
-19.8Cr-0.53Y	110.	-11.5Y ₂ O ₃	49	94
Ni-20.2Cr	.012	ZrO ₂ -17.4Y ₂ O ₃	.043	07
-19.5Cr-1.51Y	600.	-17.4Y2O3	.046	8

aNo failure.

TABLE III. - EFFECT OF YTTRICM CONCENTRATION IN NI-AI BOND COATINGS ON YTTRIA-STABILIZED ZIRCONIA THERMAL BARRIER COATING LIFE

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Bond coating	gui	Thermal barrier coating	er coating	Average number
Composition,	Thickness,	Composition,	Thickness, cm	of cycles to failure for two specimens
		990° - 250° C		
Ni-19.8Al	0.009	ZrO ₂ -7.8Y ₂ O ₃	0.042	103
-19.4Al-1.60Y	.012	-7.9Y203	.041	233
-19.3Al-0.52Y	010.	-7.9Y2O3	.042	41389
Ni-19.8Al	010.	ZrO,-11.5Y,O,	¥0.	\$
-19.4Al-1.60Y	110.	-11.5Y ₂ O ₃	.040	150
-19.3Al-0.52Y	.012	-11.5Y2O3	. 042	1256
Ni-19.8Al	010.	ZrO ₂ -17.4Y ₂ O ₃	.040	Ÿ
-19.4Al-1.60Y	010	-17.4Y203	.050	35
-19.3Al-0.52Y	110.	-17.4Y2O3	.044	448
Ni-19.8Al	.010	ZrO ₂ -24.4Y ₂ O ₃	.042	33
-19.4Al-1.60Y	110.	-24.4Y ₉ 0,	.043	29
-19.3Al-0.52Y	110.	-24.4Y2O3	.044	529
		1095° - 280° C		
Ni-19.4Al-1.60Y	0.010	ZrO,-7.8Y,O,	0.038	70
-19.3Al-0.52Y	.011	-7.8Y2O3	.038	196
Ni-19.4Al-1.60Y	010	ZrO,-11.5Y,O,	0+0	×
-19.3Al-0.52Y	.01	-11.5Y2O3	.042	160
Ni-19.4Al-1.60Y	.012	ZrO ₂ -17.4Y ₂ O ₃	.043	52
Ni-19, 4Al-1, 50Y	010	2rO ₂ -24.4Y ₂ O ₃	040	119
4				

No fatture

TABLE IV. - EFFECTS OF CHROMIUM AND ALUMINUM CONCENTRATIONS IN Ni-CF-Ai-Y BOND COATINGS ON YTTRIA-STABILIZED

[Cyclic furnace test results]

THERMAL BARRIER COATING LIFE

5
ე ₀ 082 - ₀ 06
ZrO ₂ -7.8Y ₂ C ₃
-7.8Y2O3
-7.8Y203
-7.8Y203
2r0,-11.8Y,0,
-11.8Y2O3
-11.9Y203
ZrO ₂ -17. 4Y ₂ O ₃
-17.4Y203
1095 ⁰ - 280 ⁰ €
0,-7
-7.8Y2O3
-
-7.8Y203
ZrO2-11.8Y2O3
-11.8Y203
-11.8Y ₂ O ₃
-17.4Y203

aNo failure.

TABLE V. - EFFECT OF BOND COATING THICKNESS ON YTTRIA-STABILIZED ZIRCONIA THERMAL BARRIER COATING LIFE

[Cyclic furnace test results]

Bond coating	5	Thermal barri	er coating	Average number
Composition, wt %	Thickness range, cm	Composition, wt%	Thickness, em	of cycles to failure for two specimens
	99	0 ⁰ - 280 ⁰ C		
Ni-16.4Cr-5.6Al-0.32Y	0.003-0.007	ZrO ₉ -7.8Y ₉ O ₉	0.028	1419
-0.32Y	.005008	-7.8Y2O3	.028	1529
-0.32Y	.012018	-7.8Y2O3	.032	- 1610
-1, 32Y	.018026	-7.8Y2O3	. 032	*1610
-0.32Y	.026- ,032	-7.8Y2O3	.030	~ 1 629
Ni-16, 4Cr-5, 8 Al-0, 32Y	.002007	2rO ₉ -11.8Y ₉ O ₉	.036	6 (14)
-0.32Y	.004010	-11.8Y2O3	.034	780
-0.32Y	.009015	-11.8Y2O3	. 039	1294
-0.32Y	.018027	-11.8Y2O3	.038	1119
-0.32Y	.020027	-11.8Y2O3	. 039	1132
Ni-16,8Cr-5.8Al-0.62Y	.005- ,009	ZrO ₂ -7.8Y ₂ O ₃	. 036	676
-0.62Y	.007010	-7.8Y ₂ O ₃	.040	726
-0.62Y	.014C18	-7.8Y2O3	.042	877
	109	5° - 280° C		
Ni-16, 4Cr-5, 8 Al-0, 32Y	0,005-0,008	ZrO ₂ -7.8Y ₂ O ₃	0.033	46
-0.32Y	.007010	-7.8Y ₂ O ₃	.036	60
-0.32Y	.013019	-7.87203	.038	104
Ni-16, 4Cr-5, 8 Al-0, 32Y	.004007	ZrO ₂ -11.8Y ₂ O ₃	.033	23
-0.32Y	.007010	-11.8Y ₂ O ₃	.036	35
-0.32Y	.011016	-11.8Y2O3	.038	60

^aNo failure.

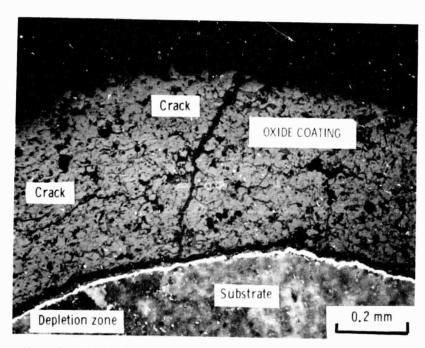
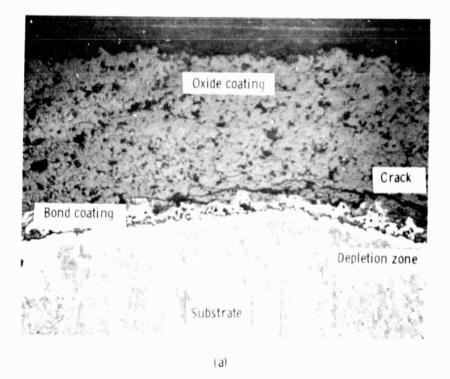


Figure 1. - Light optical photomicrograph of edge of B-1900 + Hf specimen coated with $\rm ZrO_2$ -7. 8 $\rm Y_2O_3$ after testing for 85 cycles at 990° C.





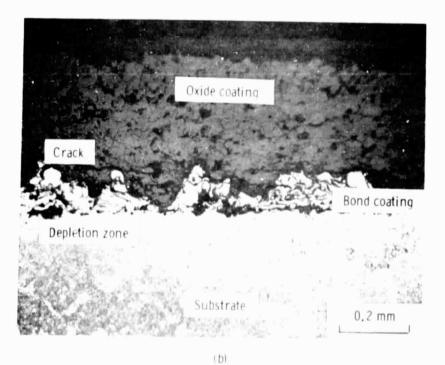


Figure 2. - Light optical photomicrograph of "(a) the edge surface of MAR-M-200 + Hf specimen coated with Ni-19, 8Cr-0, 53Y and ZrO₂-7, 8Y₂O₃ after 56 1-hour cycles at 1095⁰ C, and (b) the flat surface of MAR-M-200 + Hf specimen coated with Ni-19, 3AI-0, 52Y and ZrO₂-7, 8Y₂O₃ after 198 1-hour cycles at 1095⁰ C.

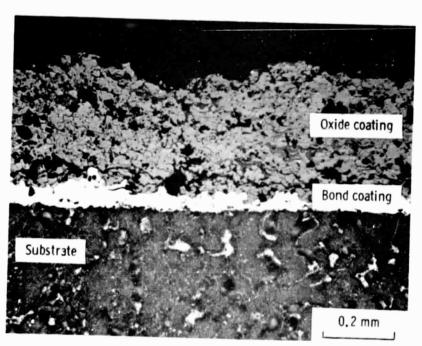
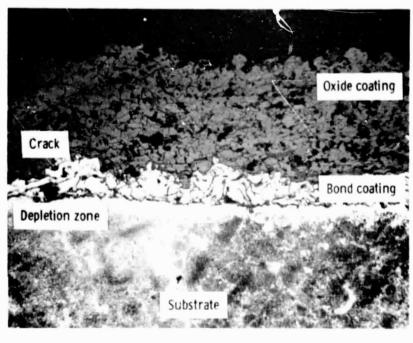


Figure 3. - Light optical photomicrograph of flat surface of B-1900 + Hf specimen coated with Ni-25. 7Cr-5. 6AI-0. 32Y and ZrO $_2$ -7. 8Y $_2$ O $_3$ after plasma spraying with no cyclic testing.



(a)

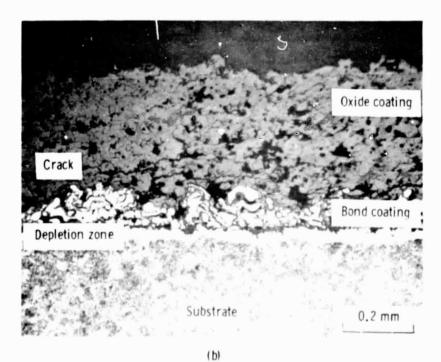


Figure 4. - Light optical photomicrograph of B-1900 + Hf specimen coated with Ni-25.7Cr-5.6Al-0.32Y and $\rm ZrO_2$ -7.8Y $_2\rm O_3$: (a) flat surface of specimen after 1500 1-hour cycles and no failure at 990° C, and (b) flat surface of specimen after 249 1-hour cycles and failure at 1095° C.